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Efficient Photocatalytic Hydrogen Evolution via Cocatalyst Loaded Al-doped SrTiO3

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1. Introduction

With population growth and technological advances, dependence on fossil energy sources is increasing, despite their limited reserves [1, 2]. The use of these fuels is accompanied by significant greenhouse gas emissions, leading to global environmental problems [3, 4]. Therefore, the scientific community is directing its efforts towards the development of renewable and clean energy sources [5, 6]. Hydrogen energy, of which water vapor is a by-product, is seen as one of the promising energy sources and requires innovative solutions to create cost-effective and environmentally friendly methods for mass production of hydrogen [7, 8, 9, 10]. One such method is the conversion of solar energy into green hydrogen through photocatalytic decomposition of water. In this context, the

development of efficient photocatalysts becomes a key challenge. Oxide semiconductors such as Sr- $TiO₃$ with a 3.2 eV forbidden bandgap have proven to be efficient photocatalysts for water splitting under UV irradiation without applying an external voltage [11]. However, high-temperature synthesis of SrTiO₃ often produces defects that reduce the photocatalysis efficiency [12]. In particular, oxygen vacancies promote the reduction of Ti^{4+} to Ti3+, which serve as recombination centers for the photogenerated electron-hole pairs [12]. Nevertheless, the $ABO₃$ -type perovskite structure has considerable flexibility, which allows improving the photocatalytic activity by controlling and minimizing defects through cation substitution in the Aand B-sites [13]. Sources emphasize that efficient separation of photogenerated electron-hole pairs can significantly enhance photocatalytic activity [14]. Fine-tuning the physicochemical properties of the photocatalyst at the atomic level, e.g., through crystal facet engineering, contributes to charge

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selective separation and increased activity. Different crystal faces can direct electrons and holes to separate regions for reduction and oxidation, as observed in photocatalysts such as BiVO₄, BiOBr, and SrTiO₃ [15]. In addition, one approach to increase the efficiency of photocatalytic reactions is the use of dual co-catalysts deposited on the photocatalyst surface. Oxidation and reduction co-catalysts play an important role by helping to form active centers and facilitating the transfer of electrons and holes to different zones on the photocatalyst surface. Studies have shown that this promotes better separation of photogenerated pairs of electrons and holes, which increases the quantum yield of the reaction [16]. While the quantum efficiency of existing photocatalysts for water splitting is usually less than 10% at UV light, T. Takata and his colleagues were able to achieve an external quantum efficiency of up to 96% at wavelengths from 350 to 360 nm [17] using a modified photocatalyst based on aluminum-doped strontium titanate. This allows the hydrogen and oxygen release reactions to be carried out separately on different crystal faces of semiconductor particles, achieving maximum quantum efficiency for water splitting and eliminating charge recombination losses. Nevertheless, studies on efficient photocatalytic total water splitting using dual co-catalysts are currently insufficient, which opens the door for further research in this area.

Previous studies in our research [18, 19, 20] extensively explored the synthesis of $SrTiO₃$ nanoparticles using a simple and cost-effective chemical precipitation method followed by calcination. This approach enables the production of $SrTiO₃$ nanoparticles with relatively high purity while maintaining good photocatalytic activity. However, modifying pure $SrTiO₃$ by aluminum doping to introduce defect engineering, followed by the photodeposition of dual co-catalysts, could further enhance the photocatalytic performance of this material, particularly for water splitting and hydrogen production.

The aim of this work is to develop and study a photocatalyst based on aluminum-doped $SrTiO₃$ with the addition of double separately photodeposited co-catalysts for efficient water splitting. A simple chemical precipitation method was employed to synthesize SrTiO $_3$ of high purity and crystallinity. The structure, morphology and photocatalytic activity of the obtained materials were studied in detail. The results demonstrate a new and simple approach to the controlled synthesis and modification of SrTiO₃, providing high efficiency of photocatalytic water splitting.

2. Materials and methods

2.1 Materials

The main reagents and chemicals were purchased from Kazakhstan and the USA. These include TiO₂ (particle size less than 25 nm, purity 99.7%, Sigma-Aldrich, USA), $Sr(NO₃)₂$ (purity \geq 98%, Sigma-Aldrich, USA), $(COOH)_{2}$ -2H₂O (purity greater than 98%, Sigma-Aldrich, USA), Al_2O_3 (particle size 50 nm, Sigma-Aldrich, USA), SrCl₂-6H₂O (Laborpharma, Kazakhstan), RhCl₃-6H₂O (Rh content 38-40%, Sigma-Aldrich, USA), Co(NO₃)₃-6H₂O (purity \geq 98%, Sigma-Aldrich, USA), and K_2 CrO₄ (purity 99.5%, Laborpharma, Kazakhstan). All chemicals were of analytical grade and did not require additional purification. Double distilled water was used for synthesis and photocatalytic experiments.

2.2 Synthesis of SrTiO₃

 $SrTiO₃$ powder was synthesized using a simple chemical precipitation method [18, 19, 20]. First, 0.12 M aqueous solution of $Sr(NO₃)$, was mixed with TiO₂ in distilled water at a Sr/Ti molar ratio of 1:1 and subjected to ultrasonic treatment for 30 minutes. A 0.4 M solution of $(COOH)_{2}$ -2H₂O was used as a reducing agent, which was added to the suspension dropwise under active stirring using a magnetic stirrer. The pH value of the suspension was then adjusted to 6–7 using 10% aqueous ammonia solution. After precipitation, the mixture was purified from excess ammonia by removing the surface liquid layer. The resulting precipitate was washed 5 times using centrifugation and dried at 60 °C for 16 hours. Finally, the product was calcined at 1100 °C in a muffle furnace in air.

2.3 Synthesis of SrTiO₃@Al

The alloying of $SrTiO₃$ powder with aluminum is carried out by the fluxing method developed by the team of Domen [17]. To synthesize SrTiO₃@ Al, powders of 1.835 g SrTiO₃, 20.39 mg Al₂O₃, and 15.85 g SrCl₂ are mixed in an agate mortar for 15 min, or longer, until homogeneous. After thorough mixing, the mixture is calcined at 1150 °C for 10 h in a muffle furnace. The resulting powder is then ultrasonically treated to completely remove residual salt, after which it is washed 5 times with hot distilled water using centrifugation and dried for 20 h at 60 °C.

2.4 Photodeposition of Rh/Cr2O3/SrTiO3@Al/CoOOH

Selective deposition of $Rh/Cr₂O₃$ and CoOOH double co-catalysts on the surface of $SrTiO₃$ -based photocatalyst was carried out by photodeposition method. First, 0.1 g of SrTiO₃ or SrTiO₃@Al powder was mixed with 50 ml of distilled water and subjected to ultrasonic treatment for 30 minutes. The resulting suspension was placed in a photochemical reactor and irradiated with ultraviolet light from a 10 W mercury lamp under vigorous stirring. After magnetic stirring was started, 50 μ l of RhCl₃-6H₂O solution (2 mg/ml concentration) was added to the mixture and irradiation was continued for 10 minutes. Then 25 μl of K_2CrO_4 solution (2 mg(Cr)/ml) was added and irradiation was continued for another 5 minutes. Finally, 25 μl of $Co(NO₃)₂$ solution (2 mg(Co)/ml) was added and irradiated for 5 minutes. The resulting samples were washed several times and dried at 70 °C overnight.

2.5 Characterization

X-ray diffraction (XRD) on a Drone-8 instrument, scanning electron microscopy (SEM) with energy dispersive X-ray spectrometer (EDX), high-resolution transmission electron microscopy (HRTEM), and X-ray photoelectron spectroscopy (XPS) were used to characterize the synthesized samples. XRD measurements were performed over an angle range of 5–70° with a step size of 0.01° and an accuracy of ±0.015°. SEM and EDX were performed on a Zeiss Crossbeam 540 at an accelerating voltage of 5–20 kV. TEM was used to analyze the structure and distribution of co-catalysts on the photocatalyst surface using a JEM-2100 LaB6 at 80 kV. XPS was performed on a Microtech Multilab 3000 VG with Mg and Al sources, and the C1s level at 284.8 eV was used for calibration. Diffuse reflectance spectra (UV-Vis DRS) were recorded on a Perkin Elmer Lambda 35 spectrophotometer in the range of 200–800 nm.

2.6 Photocatalytic reaction

To determine the photocatalytic activity of the obtained samples, 25 mg of photocatalytic sample was suspended in 50 ml of distilled water and stirred for 30 min without the addition of sacrificial agents. The photocatalytic activity was evaluated by decomposition of water followed by release of hydrogen and oxygen. A Chromos-1000 chromatograph with 3 mm columns filled with NaX and PORAPAK Q was used to analyze the released gases. The photocatalytic water decomposition process was carried out in a photochemical reactor (Shanghai Leewen Scientific Instrument Co., Ltd., China), where a 10 W high-pressure mercury lamp with wavelength $\lambda_{\text{max}} = 254$ nm was used as a radiation source. The distance between a 50 ml quartz flask and the lamp was fixed at 10 cm.

3. Results and discussion

3.1. Characterization of obtained samples

SEM and TEM techniques were used to investigate the surface morphology of the samples. In the case of SrTiO₃ calcined at 1100 °C, the formation of agglomerated nanocubes with sizes ranging from 150 to 250 nm was observed (Fig. 1a). The calcination temperature proved to be a critical factor in determining the particle size. For example, calcination in the range of 750–900 \degree C forms smaller SrTiO₃ particles, but this process is accompanied by the appearance of $S₁CO₃$ impurities [21]. Calcination at 1100 °C promotes the decomposition of carbonates, which ensures high phase purity and improves crystallinity, thereby enhancing the transfer efficiency of photogenerated charge carriers. Based on this, 1100 °C was selected as the optimum temperature for the synthesis of SrTiO₃ nanocubes. EDX analysis showed the presence of Sr, Ti and O in the structure without any extraneous impurities (Fig. 2). The doping of SrTiO₃ with aluminum nanoparticles resulted in the formation of truncated cubic particles with sizes ranging from 200 to 400 nm (Figs. 1b and 3a). Flux treatment followed by aging at 1150 \degree C promoted the broadening of $\{111\}$ facets and the reduction of {100} [22], which improved the separation and transport of photogenerated charge carriers. This increases the photocatalytic activity of $SrTiO₃$ due to more efficient water reduction with hydrogen formation and oxidation with oxygen release. Figures 3b and 3c show $SrTiO₃@Al$ samples after separate photodeposition of co-catalysts (Rh/ $Cr₂O₃$ and CoOOH) on its surface. The images show that dark formations of 5–8 nm are concentrated on some faces, while less dispersed particles of 2–5 nm are localized on others. It is assumed that $Rh/Cr₂O₃$ nanoparticles are preferentially deposited on faces {100}, while CoOOH is distributed on faces {111} and {110}. These data are in agreement with previous studies that have examined in detail the growth of co-catalysts on the surface of SrTiO₃ [17, 22]. The unique distribution structure of the co-catalysts promotes efficient separation of photogenerated charges, which in turn facilitates the photocatalytic water splitting process.

Fig. 1. SEM images of obtained (a) SrTiO₃ and (b) SrTiO₃@Al.

Fig. 2. EDX spectrum of SrTiO₃ particles.

Fig. 3. TEM images of obtained (a) SrTiO₃@Al and (b,c) obtained cocatalyst loaded SrTiO₃@Al samples.

XRD spectra were studied for a detailed analysis of the crystal structure of the synthesized samples. Figure 4a shows the X-ray diffraction spectra for SrTiO₃, SrTiO₃@Al and SrTiO₃@Al samples with deposited co-catalysts. Diagnosis of the diffraction data shows that pure $SrTiO₃$ has a cubic perovskite structure [21]. The defined peaks for $SrTiO₃$ are recorded at 2θ angles of 32.40°, 40.50°, 46.50°, 58.50° and 68.50°, which correspond to the lattice planes of the cubic phase of SrTiO₃ ((110), (111), (200), (211) and (220) as reported in JCPDS Card #35-0734 [23]. Comparison of X-ray diffraction patterns for doped and undoped $SrTiO₃$ shows a significant enhancement of peak intensities in the SrTiO₃@Al samples, indicating that aluminum doping improves the crystalline quality of $SrTiO₃$. Further, EDX mapping confirmed the presence of Ti, O, Sr and Al without any other impurities in the composition (Supplementary, Figs. S1, S2). At the same time, the presence of aluminum and co-catalysts does not change the crystal structure of SrTiO $_3$, since their peaks are not observed in the X-ray spectra. This is probably due to their low concentration, high dispersibility and low crystallinity [22].

XPS was used to analyze the changes in the chemical states and surface composition of $SrTiO₃$ samples before and after aluminum doping. The results are presented in Fig. 4 (b-d). In particular, the Ti 2p spectra (Fig. 4b) show two peaks: at 457.5 and 463.3 eV, corresponding to Ti $2p_{3/2}$ and Ti $2p_{1/2}$ states, respectively, indicating the presence of $Ti⁴⁺$ ions. The peak at 462.4 eV indicates the presence of Ti^{3+} ions in the SrTiO₃ sample [24]. After aluminum doping, $Ti³⁺$ ions completely disappear and the Ti 2p spectrum for SrTiO₃@Al contains only two peaks: 457.9 and 463.6 eV, which are associated exclusively with Ti^{4+} ions [24]. The energy level analysis for oxygen (O 1s) presented in Fig. 4 с shows two Gaussian peaks: one at 528.3 eV, corresponding to O^{2-} ions in the crystal lattice, and the other at 529.9 eV, indicating oxygen defects associated with oxygen vacancies. Aluminum doping leads to an increase in the defective oxygen ratio from 15.9 % to 54.2 %, which is consistent with data from other studies where the effect of aluminum on the photocatalytic properties of $SrTiO₃$ was investigated [25]. The optimal increase [12] in the number of oxygen defects upon aluminum doping contributes to the reduction of photogenerated carrier recombination and positively affects photocatalytic activity. The Sr 3d spectra show two peaks at 132.4 and 134.1 eV, corresponding to Sr $3d_{5/2}$ and Sr $3d_{3/2}$ states, respectively (Fig. 4d), confirming the presence of Sr^{2+} ions in SrTiO₃, which remains stable after aluminum doping. Overall, the XPS results for the synthesized samples are in agreement with other studies [24, 25].

Fig. 4. (a) XRD patterns of prepared samples; XPS spectra of the SrTiO₃ and SrTiO₃@Al samples – (b) high-resolution Ti 2p, (c) O 1s and (d) Sr 3d.

Eurasian Chemico-Technological Journal 26 (2024) 133-140

Fig. 5. (a) the UV-Vis DRS of the prepared samples and (b) the corresponding Tauc plots.

Figure 5 shows the UV-Vis absorption spectra for SrTiO₃, SrTiO₃@Al and SrTiO₃@Al with separately deposited co-catalysts in diffuse reflectance. The synthesized SrTiO₃ and SrTiO₃@Al samples exhibit light absorption up to a wavelength of 400 nm. At the same time, $SrTiO₃@Al$ with co-catalysts shows a pronounced absorption peak around 425 nm shifted to the visible spectrum. To estimate the forbidden band width of photocatalysts, the dependence of $(αhv)^2$ on photon energy was analyzed, where $α$ is the absorption coefficient, h is Planck's constant, and ν is the photon frequency [26]. The forbidden band width was 3.13 eV for SrTiO₃@Al and 3.07 eV for SrTiO₃@Al with co-catalysts. These results show that aluminum doping has no significant effect on the light absorption properties compared to pure $SrTiO₃$ (Fig. 5b). Similar results have been obtained in other studies on the doping of $SrTiO₃$ with aluminum [12]. In particular, it is noted that doping with aluminum does not change the width of the forbidden band, but mainly reduces the undesirable defects Ti³⁺, which serve as recombination centers for photogenerated charge carriers. For SrTiO₃@Al with co-catalysts, the bandgap width was slightly improved compared to SrTiO₃ and SrTiO₃@Al, indicating enhanced light absorption in the visible spectrum, especially in the violet and blue regions. However, the main advantage of co-catalysts is to reduce the recombination of photogenerated electrons and holes by improving their transport and separation [12].

3.2 Photocatalytic overall water splitting performance

Photocatalytic studies of total water splitting in the prepared samples were carried out and the results are presented in Fig. 6. The hydrogen and

Fig. 6. Line graphs of photocatalytic H_2 and O_2 production performance of (1) bare SrTiO₃, (2) SrTiO₃@Al, (3) $Rh/Cr₂O₃/STIO₃/COOOH$ and (4) $Rh/Cr₂O₃/STIO₃@Al/$ CoOOH under UV light irradiation.

oxygen production rates per hour of the used photocatalysts are in the expected stoichiometric ratio for water splitting and are shown in Fig. 6. The bare SrTiO₃ and SrTiO₃@Al samples showed very low H₂ release rates of ~0.0036 and 0.0048 mmol h^{-1} g^{-1} due to less efficient reaction sites. After loading the co-catalysts, the photocatalytic total water splitting activity of the samples increased significantly (Fig. 6). In particular, the highest activity was achieved for the $Rh/Cr₂O₃/STIO₃@Al/CoOOH sample with$ gas release rates of 11.04 and 4.69 mmol $h^{-1}g^{-1}$ for H_2 and O_2 , respectively, which significantly exceeded bare SrTiO₃ and SrTiO₃@Al by a factor of 3067 and 2300, respectively. It should be considered that the nonaluminum-doped $Rh/Cr₂O₃/STIO₃/COOOH$ photocatalyst showed a hydrogen evolution rate of 8.01 mmol h^{-1} g^{-1} under identical conditions. This confirms the results of previous studies, according to which the improved solar energy conversion

Fig. 7. Comparison of photocatalytic hydrogen production rates of (1) bare SrTiO₃, (2) SrTiO₃@Al, (3) Rh/Cr₂O₃/ SrTiO₃/CoOOH and (4) Rh/Cr₂O₃/SrTiO₃@Al/CoOOH.

ability of Al:SrTiO₃ correlates with the Al content [17, 27]. As the reaction time increased, the rate of photocatalytic hydrogen formation increased linearly and reached 55.2 and 40.06 mmol g^{-1} after 5 h in the case of $Rh/Cr_2O_3/SrTiO_3/CoOOH$ and $Rh/$ $Cr_2O_3/STIO_3@Al/CoOOH$, respectively (Fig. 7). The linearity of the hydrogen production rate at different time steps is almost the same, indicating that the catalyst performance is relatively stable during this hydrogen production process. The high photocatalytic activity of the photocatalysts with loaded co-catalysts is due to the synergistic effect of separately photodeposited $Rh/Cr₂O₃$ and CoOOH (reducing and oxidizing) co-catalysts, which promotes efficient separation and transfer of photogenerated electrons and holes. Where the photodeposited Rh/ $Cr₂O₃$ co-catalyst is responsible for the collection of

electrons with subsequent hydrogen production, whereas CoOOH releases hydrogen due to the efficient collection of holes [17, 27]. In this way, charge recombination is minimized by sequential charge transfers between the photocatalyst and co-catalysts. In addition, the results of morphology and XPS analysis show that the use of flux method for aluminum doping resulted in the separation of reduction and oxidation zones by facet engineering, as well as the suppression of defects in the form of Ti³⁺, which can serve as recombination centers.

Table shows comparisons with recent composite photocatalysts, which have relatively high results in photocatalytic water splitting. However, the photocatalyst synthesized in our laboratory significantly outperforms analogs in both H_2 and O_2 production. In addition, the above works used UV lamps with

Table. Comparison of the synthesized Cocat/STO@Al photocatalyst with the results of recent research

120 and 300 W power and used sacrificial agents to reduce recombination, which are undesirable and may cause difficulties in scaling [9].

4. Conclusion

This study successfully developed and investigated an aluminum-doped $SrTiO₃$ photocatalyst with dual, separately photo-deposited cocatalysts (Rh/ Cr_2O_3 and CoOOH) for efficient water splitting. The results demonstrated a significant enhancement in photocatalytic activity, with the $Rh/Cr_2O_3/SrTiO_3@$ Al/CoOOH sample achieving a hydrogen evolution rate of 11.04 mmol g^{-1} h⁻¹, which is substantially higher than non-modified samples. The enhanced performance is attributed to the synergistic effect of the cocatalysts, which effectively separate and transfer photogenerated charge carriers, minimizing recombination. Additionally, aluminum doping through a flux method improved crystal facet engineering and reduced defects like Ti³⁺, further contributing to the catalyst's efficiency. These findings highlight the potential of aluminum-doped SrTiO₃ and dual cocatalysts in advancing photocatalytic water splitting technologies and offer a promising approach for future research in this field.

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